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#### **Publisher's version / Version de l'éditeur:**

<https://doi.org/10.1016/j.apenergy.2016.03.115>

*Applied Energy*, 175, pp. 536-544, 2016-04-05

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Applied Energy

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# Novel hierarchical SnO<sub>2</sub> microsphere catalyst coated on gas diffusion electrode for enhancing energy efficiency of CO<sub>2</sub> reduction to formate fuel

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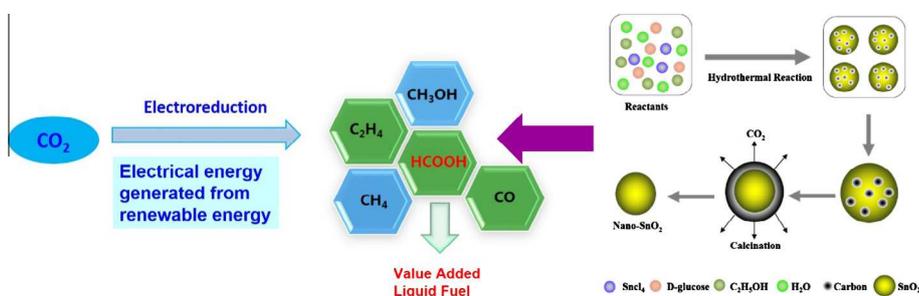
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## HIGHLIGHTS

- HMS-SnO<sub>2</sub>/GDE was developed for electrocatalytic reduction of CO<sub>2</sub> to formate fuel.
- The influence of ethanol/H<sub>2</sub>O ratio on selectivity and activity was studied by electrochemical characterization.
- A faradaic efficiency of 62.0% in a divided H-type two-compartment cell was observed at  $-1.7$  V.
- The HMS-SnO<sub>2</sub>/GDE electrode was stable over 12 h of continuous electrolysis operation.

## GRAPHICAL ABSTRACT



## ARTICLE INFO

### Article history:

Received 19 January 2016

Received in revised form 29 February 2016

Accepted 28 March 2016

Available online xxxxx

### Keywords:

Electrochemical CO<sub>2</sub> reduction

Micro/nano-SnO<sub>2</sub> particles

Formate

Faradaic efficiency

Electrode stability

## ABSTRACT

The conversion of carbon dioxide to value-added fuel using electrical energy generated intermittently from renewable energy sources is very promising in terms of energy usage reconciliation. The process converts greenhouse carbon dioxide gas to produce diverse attractive chemicals and fuels like methanol, formate, and other hydrocarbons. In this paper, the electroreduction of CO<sub>2</sub> to formate in aqueous solution is performed by using novel hierarchical tin oxide microsphere (HMS-SnO<sub>2</sub>) particles deposited over gas diffusion layer electrode (HMS-SnO<sub>2</sub>/GDE). The experiment is carried out in a divided H-type two-compartment cell with a Nafion® membrane as the diaphragm separating the cathodic and anodic compartments. The HMS-SnO<sub>2</sub> catalysts are synthesized by a facile hydrothermal self-assembled process using different ratios of ethanol to distilled water in the synthetic solution. Due to the outstanding catalytic activity and selectivity toward CO<sub>2</sub> electroreduction, SnO<sub>2</sub>-86/GDE exhibits a high Faradaic efficiency of 62% toward formate formation at  $-1.7$  V vs. SHE (Standard Hydrogen Electrode). The electrode durability is also observed with a stable current density over 12 h of continuous electrolysis operation. The superior performance is credited to the morphology- and size-controlled hierarchical structure, which may provide more active sites to accelerate the slow kinetics of CO<sub>2</sub> reduction, leading to the improved energy efficiency. During electrolysis process, KHCO<sub>3</sub> electrolyte is found to have some contribution to formate formation on the micro-structured tin oxide catalysts coated GDE electrode.

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## 1. Introduction

The emission of carbon dioxide (CO<sub>2</sub>) due to human activities has produced extra content in the atmosphere, leading to the carbon unbalance of Earth, probably resulting in negative impact on our environment through global warming. For overcoming this negative effect, several important approaches to reduce CO<sub>2</sub> emission, such as carbon capture/sequestration (CCS) and CO<sub>2</sub> conversion into useful low-carbon fuels, have been carried out in the most recent years [1]. As one of the approaches, CO<sub>2</sub> conversion to useful fuels seems to be a more attractive and promising option in terms of both increasing energy efficiency and environmental protection. Among various CO<sub>2</sub> conversion techniques, the electrochemical conversion using intermittent electricity generated from renewable energy sources has been considered one of the most feasible and innovative techniques, which can not only reduce CO<sub>2</sub>, but also produce diverse valuable fuels including formic acid, carbon monoxide, methanol, oxalic acid and hydrocarbons [2–6]. Production of useful value-added chemicals, like ethanol from CO<sub>2</sub> as an alternative to petrochemistry appears promising as it has a double advantage of reusing CO<sub>2</sub> while sparing fossil resources and avoiding CO<sub>2</sub> emissions from their use [7]. The process requires little energy input compared to their respective market values. However, due to both the extreme sluggish kinetics of CO<sub>2</sub> electroreduction and low reaction selectivity, the reduction process necessitates the use of specific metal catalysts [8]. For example, in aqueous solution, one of the main products is formic acid with high Faradaic efficiency if *sp*-group metals such as Hg, In, Sn and Pb based catalysts are employed [9].

Tin (Sn)-based materials have been considered to be promising electrocatalysts for CO<sub>2</sub> reduction to produce small fuels (e.g. formate, methanol or hydrocarbons) because of their low cost, wide availability and reasonable overall Faradaic efficiency [10]. Normally, the Sn metal based electrodes could also have high selectivity in the production of formate in aqueous electrolytes. This is particularly attractive because the formate can be used as a fuel for direct formate fuel cells [11] and also as the optimal hydrogen carrier [12]. Currently the formate is produced by hydrolysis of methyl formate and hydrolysis of formamide, with negative environmental impact and relative expensive production costs. Developing an efficient electrocatalytic process for formate production could reduce the costs and enhance its use in fuel cells and related applications [13]. These formate products may thus offset costs associated with carbon management. Besides, the introduction of such carbonaceous products to markets might significantly improve the world economy by lowering production costs of several intermediates, providing massive and cheap production of some final products [7]. However, as realized, the deactivation of Sn metal based electrodes during CO<sub>2</sub> reduction is very fast, and the reduction reaction on these electrodes requires at least ~860 mV of overpotential at a current density of 4–5 mA cm<sup>-2</sup> in an aqueous solution saturated with 1 atm CO<sub>2</sub> [14]. To overcome these challenges, several approaches have been explored using different structures of Sn-based electrodes, such as Sn particles (shots and granules), Sn foil, or Sn-based gas diffusion electrodes [15–20].

Most recently, Chen and Kanan [21] reported a SnO<sub>x</sub>/Sn composite electrode could participate in the CO<sub>2</sub> reduction pathway by providing chemical functionality that might stabilize the incipient negative charge on CO<sub>2</sub> (i.e., CO<sub>2</sub><sup>-</sup>) or by mediating the electron transfer directly, therefore the Faradaic efficiency of ~58% for formate formation was obtained on such a SnO<sub>x</sub>/Sn electrode, which was much higher than that of ~19% observed on pure Sn foil electrode. Zhou et al. [10] found that if the electrode was only composed of SnO<sub>2</sub> layer with a thickness of ~3.5 nm, the highest Faradaic efficiency (64%) towards formate formation could be observed at -1.2 V. Due to the controlled structure and morphology, the metal/metal oxides nanoparticle surfaces may provide

abundant undercoordinated sites, which are more likely to be the active sites for CO<sub>2</sub> reduction [22–24]. All above previous works reported single Sn metal or SnO<sub>x</sub>/Sn electrode, which suggested the application of SnO<sub>x</sub> on CO<sub>2</sub> electroreduction showed very interesting and promising results. This insight has indicated that SnO<sub>x</sub> catalyst exhibits greatly enhanced CO<sub>2</sub> reduction activity relative to a typical Sn electrode. Furthermore, as this material is stable in most organic and aqueous solutions, it is a promising candidate for CO<sub>2</sub> electroreduction.

In this work, the operation of a continuous electroreduction process of CO<sub>2</sub> to formate in aqueous solutions is studied by using electrodes with hierarchical tin oxide microsphere (HMS-SnO<sub>2</sub>), which are a new series of morphology and size controlled HMS-SnO<sub>2</sub> catalysts with high catalytic efficiency for CO<sub>2</sub> electroreduction. The catalyst structures are induced by a facile hydrothermal self-assembled process through changing different ratios of ethanol to distilled water in the synthetic solution. The advantages of this synthesis method are a decrease in the electrochemical potential at which electrochemical reductions occur and an increase in the selectivity of the process. To maximize surface area, all the synthesized SnO<sub>2</sub> microsphere catalysts are coated on the gas diffusion layer (GDL) to form 3D porous structures for facilitating the mass transportation from the gas–liquid interface to the catalyst surface [25–27]. In this way, easy diffusion path lengths can be achieved for substrates to access, leading to faster kinetics. Furthermore, the evolution of byproduct H<sub>2</sub>, which could impede the reduction reaction between CO<sub>2</sub> and electrode, could also be inhibited effectively [28]. Therefore, the aim of this work is to study the electrode (SnO<sub>2</sub> coated on the GDL) morphology and size effects on catalytic activity, product selectivity, Faradaic efficiency and energy efficiency. It is also studied systematically using techniques such as cyclic voltammetry, linear sweep voltammetry, and ion chromatography. Electrochemical measurements show the catalysts can have outstanding catalytic activity, selectivity and good stability toward CO<sub>2</sub> electroreduction. These electrochemical tests can help to provide a more detailed understanding of the process and the influence of these parameters on the resulting reactivity. During CO<sub>2</sub> reduction process, catalysts hold the key to successful industrial applications. Therefore, this work could effectively increase the number of industrial applications by improving the CO<sub>2</sub> electroreduction reactions for commercialization and should be an integral part of research and development for carbon management and sustainable development.

In our work, with 1–3 μm large SnO<sub>2</sub> microspheres composed of sub-structured nanoparticles (20–40 nm), a maximum Faradaic efficiency of >62% for formate production is achieved. These catalysts are very stable during electrolysis and can continue producing formate for at least 12 h. Possible catalytic mechanism for these SnO<sub>2</sub> nanoparticle coated electrodes is discussed based on the measurements and analysis of scanning electron microscopy, crystal-phase X-ray diffraction pattern, high-resolution transmission electron microscopy and selected area electron diffraction analyses, as well as X-ray photoelectron spectroscopy. For further understanding, the chemical compositions of the electrode materials before and after a long time electrolysis are also analyzed. In addition, the effect of KHCO<sub>3</sub> electrolyte on the CO<sub>2</sub> reduction is also investigated using both the N<sub>2</sub>-saturated KHCO<sub>3</sub> solution and CO<sub>2</sub>-saturated pure H<sub>2</sub>O for comparison.

## 2. Experimental

### 2.1. Catalyst synthesis

The SnO<sub>2</sub> microsphere catalysts were synthesized by a facile hydrothermal self-assembled process, where SnCl<sub>4</sub> and D-glucose



**Scheme 1.** Representative illustration of the formation of HMS-SnO<sub>2</sub> in solvothermal synthesis.

monohydrate were used as the precursors, as illustrated in Scheme 1. For a typical synthesis, 4 mmol of SnCl<sub>4</sub> and 10 mmol of D-glucose monohydrate were dissolved in a mixture of distilled water and ethanol (35 mL) under stirring till forming a transparent colorless solution. Then this mixture solution was transferred into a 100 mL Teflon-lined stainless steel autoclave, which was sealed and maintained at 180 °C for 24 h before it was cooled down in air. The sediment with a black color was collected and washed by ethanol/water several times, and was dried in a vacuum oven at 60 °C for 5 h. The final powder was calcined at 550 °C for 5 h in air, during which the black sediment was gradually turned into a white one, indicating the successful removal of carbon by oxidation in air. In this way, the SnO<sub>2</sub> microspheres with controlled size and morphology were obtained just by changing the volume

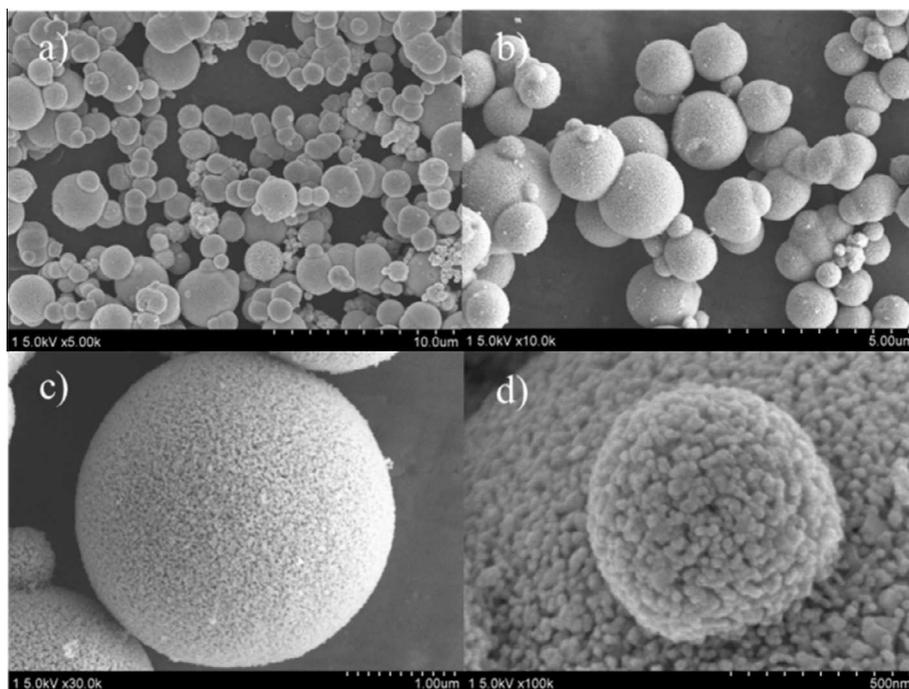
ratio of ethanol to distilled water as shown in Fig. 1 (30:5) and Fig. 3 ((a) 10.5:24.5, (b) 24.5:10.5, (c) 30:5, (d) 35:0).

## 2.2. Electrode preparation and electrochemical test

For working electrode preparation, the SnO<sub>2</sub> microsphere catalysts synthesized above were used to prepare catalyst inks which were then coated onto the carbon paper sheet (CPS) (Toray, TGP-H-090) to form gas diffusion electrodes (GDEs). In a typical preparation, as-prepared SnO<sub>2</sub> catalyst (3 mg cm<sup>-2</sup> loading) was dispersed in a mixture of 100 mg of 5 wt.% Nafion solution and 1.4 mL of 99.7 wt.% isopropyl alcohol (Sinopharm Chemical Reagent Co.) to form catalyst ink [29], which was then coated on CPS (4 cm<sup>2</sup>) to form a SnO<sub>2</sub> microsphere catalyst coated gas-diffusion electrode (abbreviated as SnO<sub>2</sub>/GDE).

For electrochemical characterization, a standard H-type cell (Aldrich Nafion<sup>®</sup>117) equipped with both gas inlet and outlet, which allow the passage of either N<sub>2</sub> (99.99%) or CO<sub>2</sub> (99.99%) through the solution, was used for measuring both catalyst property and CO<sub>2</sub> reduction performance. In the measurements, SnO<sub>2</sub>/GDE was used as the working electrode and a piece of Nafion<sup>®</sup>117 cation exchange membrane (H<sup>+</sup> form) as a separator, platinum foil electrode as the counter electrode and a saturated calomel electrode (SCE) as the reference electrode, respectively. Prior to all experiments, the solution was purged with N<sub>2</sub> for at least 30 min to remove all dissolved air. For CO<sub>2</sub> reduction measurements, the solution was bubbled with 1.0 atm CO<sub>2</sub> gas (99.99%) for 30 min before measurements. All electrochemical measurement was performed on a CH Instrument 660E.

The electrocatalytic activity and kinetics of SnO<sub>2</sub>/GDE working electrode were tested using both cyclic voltammetry (CV) and linear sweep voltammetry (LSV) at scan rates of 50 mV s<sup>-1</sup> and 5 mV s<sup>-1</sup>, respectively, and in a potential range of 1.0~–1.6 V vs. SCE. The controlled potential electrolysis was carried out using a CHI660E electrochemical analyzer in the same standard H-type cell. Before each electrolysis process, the electrolyte was bubbled with CO<sub>2</sub> for at least 30 min, and CO<sub>2</sub> gas was continuously



**Fig. 1.** SEM images of HMS-SnO<sub>2</sub> aggregates at different magnifications, i.e., (a) (×5000), (b) (×10,000), (c) (×30,000), and (d) (×100,000). The synthetic solution with ethanol/water ratio to be 30:5.

bubbled to maintain the saturation during the following electrolysis. For the potentiostatic measurements, a constant potential of  $-1.70$  V vs. SHE was imposed for 12 h for stability test, where the electrolysis current was continuously recorded. For soluble product measurements, the electrolysis time was controlled for 60 min. All these tests were carried out at ambient temperature and pressure.

### 2.3. Physical characterization and product analysis

The morphology of as-prepared SnO<sub>2</sub> microsphere catalysts was characterized by Zeiss ultra plus thermal field emission SEM instrument (Carl Zeiss SMTAG, Germany). The crystal-phase X-ray diffraction (XRD) patterns of SnO<sub>2</sub> catalysts were obtained using a Philips PW3830 X-ray diffractometer equipped with CuK $\alpha$  radiation ( $\lambda = 0.15406$  nm). The intensity data was collected at 25 °C in the 2 $\theta$  range from 0° to 90° with a scan rate of 1.20 min<sup>-1</sup>. High-resolution transmission electron microscopy (HR-TEM) and selected area electron diffraction (SAED) analyses were performed with a high-resolution Hitachi JEM-2100F operating at 200 kV to obtain the information of the average particle size and the crystal properties of catalysts. For chemical composition measurement after long time electrolysis, the SnO<sub>2</sub> microsphere catalysts were analyzed by the X-ray photoelectron spectroscopy (XPS) measurements on a Kratos AXIS UltraDLD electron spectrometer with Al K X-ray anode source ( $h\nu = 1486.6$  eV) at 250 W and 14.0 kV. For the analysis of the reduction soluble products, before product measurements, the solution after CO<sub>2</sub> reduction was diluted 50 times or 100 times (which is convenient to formate being detected in mobile phase), then it was filtered with filter membrane (0.22  $\mu$ m) and the products in the electrolyte were directly analyzed by ion chromatography (ICS-90, Dionex, USA) using an AS144 mm  $\times$  250 mm separation column at a flow rate of 1 mL/min. In this analysis, the mobile phase was a mixed aqueous solution of Na<sub>2</sub>CO<sub>3</sub> (3.5 mM) and NaHCO<sub>3</sub> (1.0 mM), and a H<sub>2</sub>SO<sub>4</sub> (20 mM) aqueous solution was used as a regenerative liquid.

## 3. Results and discussion

### 3.1. Surface structure of SnO<sub>2</sub> microsphere catalysts

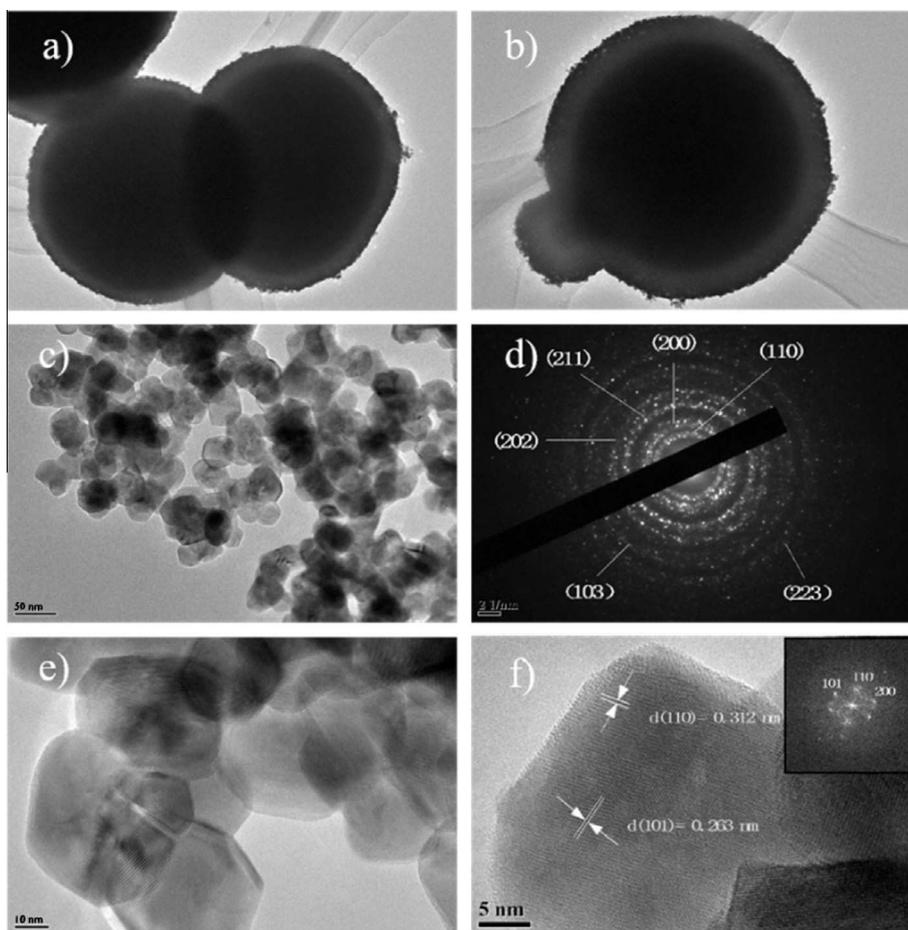
As typical catalyst sample, Fig. 1(a)–(c) shows three different magnifications of SEM images of SnO<sub>2</sub>-86 catalyst, which was prepared using a synthetic solution with ethanol/water ratio of 30:5. It can be observed that the SnO<sub>2</sub>-86 catalyst shows a clear and uniform sphere-like structure with 1–3  $\mu$ m in overall dimension (Fig. 1a and b). The high magnification SEM image of Fig. 1c indicates a large sphere size in diameter of 3  $\mu$ m. Further from the high magnification SEM image as indicated by Fig. 1d, one spherical particle of 680 nm can be clearly observed which parasitized on large microsphere particle surface. All these large spherical particles indicate a clear 3-dimensional hierarchical structure, which are comprised of the secondary structures entirely by aggregated small primary SnO<sub>2</sub> nanoparticles in the size of 20–40 nm in diameter. This special morphology structure may greatly influences the catalytic activity of the catalysts for CO<sub>2</sub> electroreduction, which would be demonstrated in the following section.

The formation mechanism can be described as schematically illustrated in Scheme 1. In order to determine the dominating factors in the process, a series of experiments were carried out to understand the formation of SnO<sub>2</sub> nanoparticle aggregates. The interesting microstructure of an as-synthesized large hierarchical microsphere SnO<sub>2</sub> (HMS-SnO<sub>2</sub>) can be further corroborated by TEM images, as displayed in Fig. 2a and b. In agreement with the above SEM observations (Fig. 1c and d), a high uniformity of the

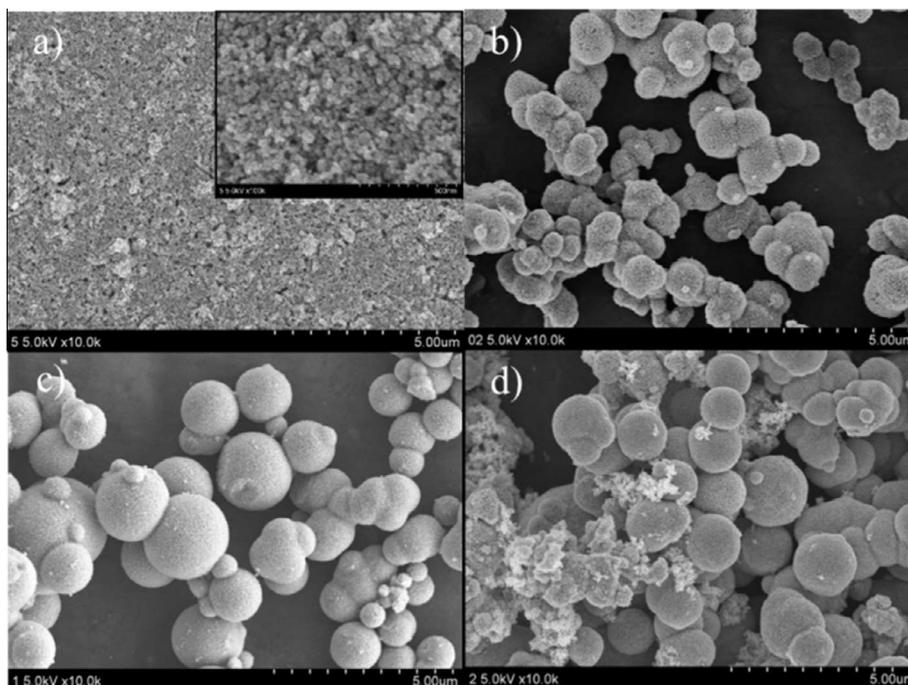
microspheres can be seen from the TEM images. A high-magnification TEM image of nanoparticles further shows the microcosmic characteristics of microsphere SnO<sub>2</sub> as shown in Fig. 2c. The nanoparticles display a similar polygon structure of approximately 20–40 nm. The TEM bright-field imaging combined with selected-area electron diffraction (SAED) reveals the fine microstructure of the HMS-SnO<sub>2</sub>, each wire being a monocrystal with a tetragonal rutile structure, as shown in Fig. 2d [30]. The corresponding ring-like SAED pattern (Fig. 2d) indicates the nanostructures are crystalline, and the diffraction rings from inside to outside can be indexed to (110), (200), (211), (202), (103) and (223) planes of SnO<sub>2</sub> nanocrystals with hexagonal symmetry, respectively. High-resolution (HR) imaging was combined with SAED to investigate the nanoparticles growth direction. As shown in Fig. 2e, the HR-TEM image of nanoparticles shows a clear shape similar to a polygon structure. Fig. 2f further shows a high-magnification image of HR-TEM, indicating that the nanoparticles are exposed two growth directions of (110) and (101) facets. Fourier transforms (FFTs) of the HR-TEM image (insets) show three facets, (110), (101) and (200), which are in accordance with the XRD results (Fig. S1). Meanwhile, the FFTs combined with macroscopic XRD further confirms the rutile structure.

In order to determine the key factors in the microsphere synthesis, the volume ratio of ethanol to distilled water were changed to ascertain the formation of SnO<sub>2</sub> nanoparticles aggregates. It is found the ethanol/water ratio plays an important role in the formation of microspheres morphology. Fig. 3(a)–(d) shows the SEM images of SnO<sub>2</sub> as-prepared from different ratios of ethanol to distilled water as solvent, i.e., SnO<sub>2</sub>-30 and SnO<sub>2</sub>-70, SnO<sub>2</sub>-86 and SnO<sub>2</sub>-100, respectively. Fig. 3a shows that when the ratio is 30%, only a plane composed of some tight SnO<sub>2</sub> nanoparticles could be formed, as can be seen in the insert image of Fig. 3(a). When the ethanol content in the mixed solvent is increased to 70% (i.e., SnO<sub>2</sub>-70), the morphology of SnO<sub>2</sub> begins to change largely to some microspheric aggregates (Fig. 3b), the size of SnO<sub>2</sub> particles is increased dramatically up to 1–1.5  $\mu$ m with a large microsphere structure. However, there still exists numerous microspheres accumulated. The distinguishable HMS-SnO<sub>2</sub> particles can only be produced when the ethanol content is larger than 80% (i.e., SnO<sub>2</sub>-86 and SnO<sub>2</sub>-100) (Fig. 3c and d). However, in a pure ethanol solution (i.e., SnO<sub>2</sub>-100), the interior of the microsphere structures begins to collapse and becomes small aggregated nanoparticle of ca. 20–30 nm, which is formed surrounding the SnO<sub>2</sub> microsphere (Fig. 3d).

The crystallinity and phase purity of as-prepared SnO<sub>2</sub> aggregates are independently confirmed by XRD (Fig. S1(a–d)). All the peaks in the XRD pattern in Fig. S1 could be indexed to crystalline SnO<sub>2</sub> by comparing with JCPDS card No. 41–1445, indicating it has good phase purity, and the crystalline SnO<sub>2</sub> nanoparticles are small in size. All SnO<sub>2</sub> aggregates have the similar diffractions, although SnO<sub>2</sub>-86 sample shows four additional small peaks (Fig. S1(c)). In comparison with other SnO<sub>2</sub> samples, the SnO<sub>2</sub>-86 sample has a well-defined morphology of hierarchical microsphere structure, as shown in Fig. 1c and d, which may be responsible for these additional four peaks. It can also be seen the crystallinity of all SnO<sub>2</sub> aggregates becomes pronounced as indicated by the increased intensities of the four diffraction peaks of (110), (101), (200) and (211), which can be assigned to tetragonal SnO<sub>2</sub>. The main diffraction (110), (101), (200), and (211) peaks are relatively broad, indicating that the nanoparticles are composed of SnO<sub>2</sub> nano-crystalline. The crystallinity can be calculated to be 71.7% and the crystalline grain sizes of these four crystal surfaces are 14.8, 17.1, 15.4 and 15.7 nm, respectively. A panoramic view reveals that SnO<sub>2</sub>-86 is entirely consist of uniform microspheres without impurity particles or aggregates, as clearly observed in the SEM images (Fig. 1).



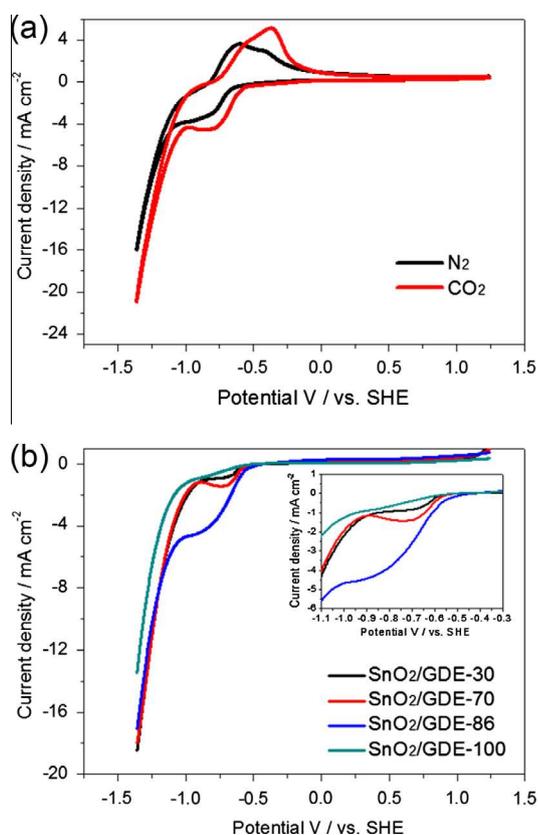
**Fig. 2.** (a, b) Low-magnification TEM images of as-prepared HMS-SnO<sub>2</sub>, (c, e, f) Enlarged TEM images of HMS-SnO<sub>2</sub> at different magnifications with (f) insert FFT pattern, (d) The selected area electron diffraction pattern (SAED) of HMS-SnO<sub>2</sub>. The synthetic solution with ethanol/water ratio to be 30:5.



**Fig. 3.** SEM images of solvothermal products prepared from different ratios of ethanol to distilled water as solvent. (a) SnO<sub>2</sub>-30, (b) SnO<sub>2</sub>-70, (c) SnO<sub>2</sub>-86 and (d) SnO<sub>2</sub>-100, respectively.

### 3.2. Electrocatalytic performance toward CO<sub>2</sub> reduction

To evaluate the CO<sub>2</sub> reduction activity of the HMS-SnO<sub>2</sub> electrode prepared from SnO<sub>2</sub>/GDE-86, cyclic voltammetric (CV) measurements were carried out in both N<sub>2</sub>- and CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> solution, separately, as shown in Fig. 4a. The anodic peaks between -0.3 V and -0.8 V and the cathodic peak between -0.7 V and -1.0 V in CV curves can be attributed to the formation and reduction of tin oxides in basic media, respectively [31]. The SnO<sub>2</sub>/GDE-86 electrode exhibits a less negative onset potential at -0.48 V vs. SHE, which is 100 mV more positive than that of under N<sub>2</sub>. At a high overpotential of -1.36 V, a sharp increase in the current densities can be observed under both N<sub>2</sub> and CO<sub>2</sub>. In the blank experiment, this increase is due to the hydrogen evolution reaction (HER), while the higher current density in the presence of CO<sub>2</sub> can be ascribed to the improved kinetics of CO<sub>2</sub> reduction, and the enhanced current represents its reduction to formate. In order to deeply understand the effect of microsphere morphology on the CO<sub>2</sub> electrocatalytic performance, Fig. 4b presents the linear sweep voltammetric (LSV) scans in CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> solution on four electrodes, i.e., SnO<sub>2</sub>/GDE-30, SnO<sub>2</sub>/GDE-70, SnO<sub>2</sub>/GDE-86 and SnO<sub>2</sub>/GDE-100, respectively, where the SnO<sub>2</sub>/GDE-86 electrode can give the earliest onset potential (about -0.49 V vs. SHE) compared to other three electrodes. As shown in Fig. 4b, the catalytic activity is increased with increasing the content of ethanol in the synthetic solution from 30% to 70%, as indicated by positively moving of onset potential. The maximum current density reaches 18 mA cm<sup>-2</sup> for SnO<sub>2</sub>/GDE-70 electrode. Note that when the content of ethanol in the synthetic solution is continued to be increased, the size of microspheres is increased, and the morphology of SnO<sub>2</sub> is changed to an independent microsphere, as shown

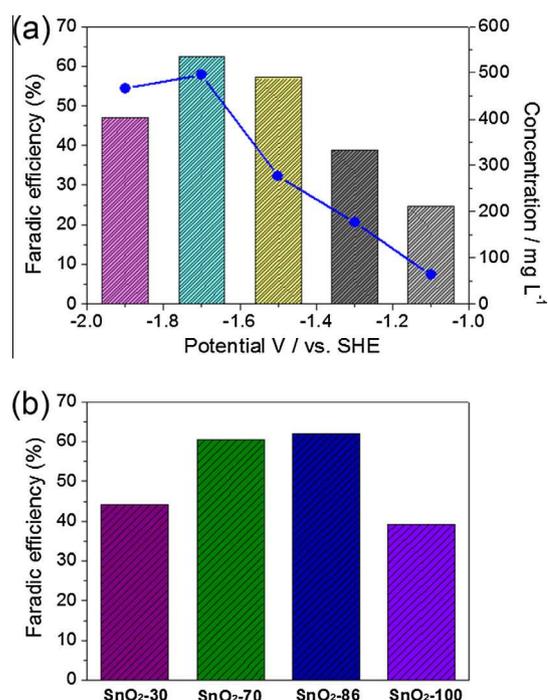


**Fig. 4.** (a) cyclic voltammetry (CV) curves of SnO<sub>2</sub>/GDE-86 in N<sub>2</sub>- and CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> electrolytes, and (b) Linear sweep voltammetric (LSV) of SnO<sub>2</sub>/GDE-30, SnO<sub>2</sub>/GDE-70, SnO<sub>2</sub>/GDE-86 and SnO<sub>2</sub>/GDE-100, scanning at 5 mV/s in CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub>.

in Fig. 3(b) and (c). Such changes may contribute to the catalytic activity for CO<sub>2</sub> electroreduction, probably leading to the enhanced energy efficiency. It can be seen the SnO<sub>2</sub>/GDE-86 electrode, which possesses HMS-SnO<sub>2</sub> morphology on the electrode surface, gives the most positive onset potential shift of the 80 mV compared to SnO<sub>2</sub>/GDE-30, and 60 mV compared to SnO<sub>2</sub>/GDE-70. In addition, on the cathodic end of the curves, the maximum current density reaches to 17 mA cm<sup>-2</sup> at the same time. When the content of ethanol in the synthetic solution is further increased (SnO<sub>2</sub>/GDE-100 electrode), the performance for CO<sub>2</sub> reduction is decreased sharply with a current density of 13.4 mA cm<sup>-2</sup> maintained, and a more negative onset potential of -0.56 V under the same conditions. As described in Fig. 3d, when the content of ethanol is increased to 100% while maintaining other synthesis condition unchanged, some microspheres structure is destroyed and the solid nanoparticle aggregates are appeared around SnO<sub>2</sub> microspheres. The decreased catalytic activity of SnO<sub>2</sub>/GDE-100 might be partly caused by the destroyed HMS-SnO<sub>2</sub> structure [32].

### 3.3. Faradaic efficiency for electrochemical reduction of CO<sub>2</sub> to formate

Controlled potential electrolysis was performed to investigate the effect of applied potentials on Faradaic efficiencies for formate production at HMS-SnO<sub>2</sub> electrode. Electrolysis experiments were controlled in a range from -1.1 to -1.9 V vs. SHE in CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> solution, and the results are shown in Fig. 5a. The SnO<sub>2</sub>/GDE-86 electrode with the best catalytic activity is selected as the target electrode. As reported in earlier reports [33], formate was observed as the dominant soluble product at negative potentials. In our experiments, it is observed that the production of formate on the HMS-SnO<sub>2</sub> electrode is strongly dependent on the electrode potential. After -1.1 V, the formate is started to be produced, and the amount of formate reached to 62.8 mg L<sup>-1</sup>. With further decreasing the cathode potential from -1.1 to -1.9 V vs. SHE, both the current density and the amount of formate are increased. As shown in Fig. 5a, the SnO<sub>2</sub>/GDE-86



**Fig. 5.** (a) Formate production and faradaic efficiency of SnO<sub>2</sub>/GDE-86 electrode with different potentials on different electrodes and (b) Faradaic efficiency of SnO<sub>2</sub> prepared from different proportions of ethanol to distilled water as solvent. Electrolysis time: 1 h.

electrode shows formate production range from 60 to 500 mg L<sup>-1</sup>, reaching to a maximum at -1.7 V vs. SHE, and it is almost 6.5 times of that at a potential of -1.1 V. When the electrolysis potential is further decreased, the trend of the formate production shows a slight drop. Furthermore, Fig. 5a shows the Faradaic efficiencies for formate production as a function of electrode potential during the electrolysis process. The trends shown in Fig. 5a for the Faradaic efficiency as a function of potential are in an agreement with those reported for pure Sn metal electrodes [11,34–36]. It can be seen that the Faradaic efficiency for formate is increased when increasing the cathode potential from -1.1 V to -1.7 V, where it reaches to the maximum with a Faradaic efficiency of 62% at current density of 12.5 mA cm<sup>-2</sup>. This value is a relatively high Faradaic efficiency when compared to previous reports as seen in Table S2. As the potential is decreased further, the Faradaic efficiency begins to drop. This may be caused by hydrogen evolution at very negative potentials [28,37,38]. This hydrogen evolution could compete with CO<sub>2</sub> reduction under these conditions, which can be observed by a large amount of hydrogen bubbles on the electrode surface. In order to deeply understand the effects of proportions of the ethanol to distilled water as a solvent on the catalysts' selectivity, the produced formate was determined and analyzed by ion chromatography (IC), after applying a constant potential of -1.7 V vs. SHE in a CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> electrolyte for 60 min. From Fig. 5b, it can be seen the highest Faradaic efficiency obtained in this work can be reached up to 62% for SnO<sub>2</sub>/GDE-86 electrode. It is noted for the SnO<sub>2</sub>/GDE-70 electrode, the Faradaic efficiency shows a similar value as SnO<sub>2</sub>/GDE-86, indicating the integrated microsphere structure has a large effect on the selectivity of SnO<sub>2</sub> catalyst. When a lower content (30%) of ethanol is present in the synthetic solution, the Faradaic efficiency is sharply decreased, where there is no formation of HMS-SnO<sub>2</sub> structure. However, when the amount of ethanol is extremely high such as 100%, the Faradaic efficiency for producing formate also shows some drop, to where some HMS-SnO<sub>2</sub> structure is broken into small aggregated particles. This further demonstrates the Faradaic efficiency for CO<sub>2</sub> electroreduction strongly depends on the surface structure such as morphology and size control of the catalysts as prepared [22–24]. For a fair comparison, all detailed results of Faradaic efficiency and concentration of formate are presented in Table S1.

#### 3.4. Effect of KHCO<sub>3</sub> electrolyte for electroreduction CO<sub>2</sub>

It was reported the KHCO<sub>3</sub> electrolyte solution in the absence of CO<sub>2</sub> had some effect on formate production, and the Faradaic efficiency toward formate production could be 2.7% with Sn electrode [37]. Here, we did some measurements for investigating the effect of purging gas, CO<sub>2</sub> or N<sub>2</sub>, and HCO<sub>3</sub><sup>-</sup> on the Faradaic efficiency toward formate production on SnO<sub>2</sub>-86/GDE electrode, and the results are shown in Fig. 6. In 0.5 M KHCO<sub>3</sub> electrolyte with nitrogen bubbling, some formate was detected with IC. The Faradaic efficiency of formate reached to 2.4% and the concentration reached to 13.2 mg L<sup>-1</sup>. It should be pointed out that some amount of formate was also observed when purging with nitrogen in 0.5 M KHCO<sub>3</sub> electrolyte solution, which could be attributed to the electrochemical reduction of CO<sub>2</sub> derived from HCO<sub>3</sub><sup>-</sup> due to the homogenous acid-base reaction (1), the produced CO<sub>2</sub> is then reduced through Reaction (2) to produce formate along with the hydrogen evolution reaction (HER, Reaction (3)) [29,39].

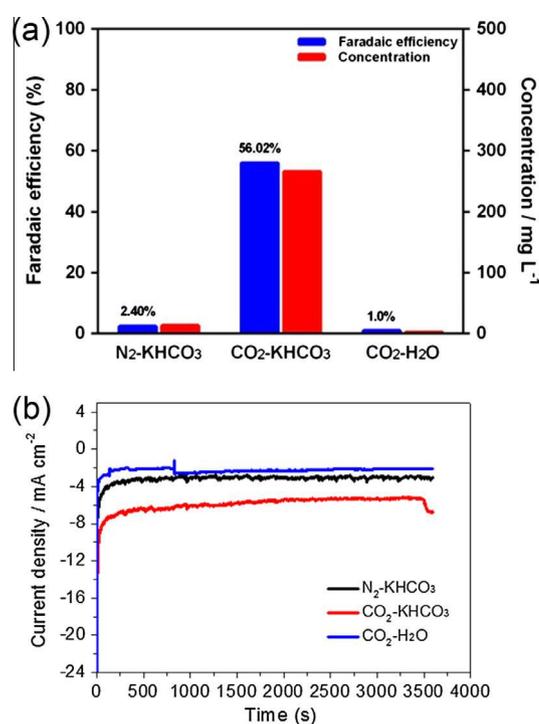
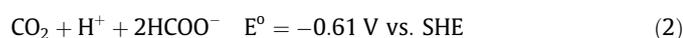


Fig. 6. (a) Faradaic efficiencies and concentrations of formate production and (b) the current density with electrolysis time for three electrolytes, i.e., N<sub>2</sub>-KHCO<sub>3</sub>, CO<sub>2</sub>-KHCO<sub>3</sub> and CO<sub>2</sub>-H<sub>2</sub>O at -1.5 V vs. SHE. Electrolysis time: 1 h.

As shown in Fig. 6a, upon the solution is saturated with CO<sub>2</sub>, the Faradaic efficiency toward formate production is increased to 56.0% (at -1.5 V vs. SHE), and the formate production is reached to 260 mg L<sup>-1</sup>, which is almost 20 times of that under N<sub>2</sub>. Moreover, the current density for the reduction reaction is increased by 1.7 times when purging with CO<sub>2</sub> compared to N<sub>2</sub>. A similar improvements in the Faradaic efficiency, from 2.4% to 56.0%, and for the current density, from 3.1 mA cm<sup>-2</sup> to 5.4 mA cm<sup>-2</sup>, can be observed when exchanging N<sub>2</sub> for CO<sub>2</sub> purging in 0.5 M KHCO<sub>3</sub> solution. When replacing KHCO<sub>3</sub> with H<sub>2</sub>O for CO<sub>2</sub> purging, the Faradaic efficiency of formate is as low as 1.0%, and the corresponding formate production is only reached to 2.2 mg L<sup>-1</sup>, which is lower than KHCO<sub>3</sub> electrolyte with N<sub>2</sub> purging, and the current density is as low as 2.0 mA cm<sup>-2</sup>, as shown in Fig. 6b. When using H<sub>2</sub>O as the electrolyte, it cannot provide sufficient ions for the mass transfer diffusion on the surface of electrode, where there are only a little of the hydrogen ions reacted with CO<sub>2</sub> to form formate. When the electrolyte is pure water saturated with CO<sub>2</sub>, the equilibrium of CO<sub>2(aq)}/HCO<sub>3</sub><sup>-</sup>/CO<sub>3</sub><sup>2-</sup> can give 99.7% of the dissolved CO<sub>2</sub> in the form of liquid CO<sub>2</sub>, and only 0.3% in the form of carbonate. In this case, the concentration of HCO<sub>3</sub><sup>-</sup> is too low to provide conductivity of the solution for formate production. On the contrary, in KHCO<sub>3</sub> electrolyte saturated with nitrogen, there are large amounts of HCO<sub>3</sub><sup>-</sup> and small amounts of CO<sub>3</sub><sup>2-</sup>, which does not have liquid CO<sub>2</sub>, leading to a trace amount of formate production by the equilibrium between HCO<sub>3</sub><sup>-</sup> and CO<sub>2</sub> [40,41]. When replacing N<sub>2</sub> with CO<sub>2</sub>, large amounts of liquid CO<sub>2</sub> and HCO<sub>3</sub><sup>-</sup> are saturated in the electrolyte solution, which can provide sufficient CO<sub>2</sub> to the electrode for reduction. KHCO<sub>3</sub> solution supplies the most availability of reactants for CO<sub>2</sub> reduction leading to a higher Faradaic efficiency at more negative potentials [16]. The highest reduction current density can be observed in 0.5 M KHCO<sub>3</sub> and is attributed to the equilibrium between HCO<sub>3</sub><sup>-</sup> and CO<sub>2</sub>, which provides sufficient local dissolved CO<sub>2</sub> to the interface between SnO<sub>2</sub>/GDE-86 electrode and electrolyte for the reaction, benefiting from the large hierarchical SnO<sub>2</sub> microsphere structure.</sub>

### 3.5. Stability of HMS-SnO<sub>2</sub> modified GDE for CO<sub>2</sub> electroreduction

To test the catalyst's stability, SnO<sub>2</sub>-86/GDE electrode with a hierarchical microsphere structure was used, and the electrochemical reduction of CO<sub>2</sub> was controlled at  $-1.7$  vs. SHE. Fig. 7a shows the current density as a function of electrolysis time. It can be seen that the total reduction current density as high as  $12.2 \text{ mA cm}^{-2}$  can be achieved in  $0.5 \text{ M KHCO}_3$ . The current density on the electrode is decreased sharply in the initial stage of the reaction, indicating that there is an intermediate product generation along with the CO<sub>2</sub> reduction reaction occurring on the electrode at the initial stage. However, after that the current density remains almost unchanged. With the extension of time, the reduction reaches to a steady state in the electrolyte, implying that the nature of the cathodic reaction is not changed. In order to further demonstrate this point, we have taken a comparison of XPS spectra of the cathode before and after electrolysis. The atomic percentages of Sn

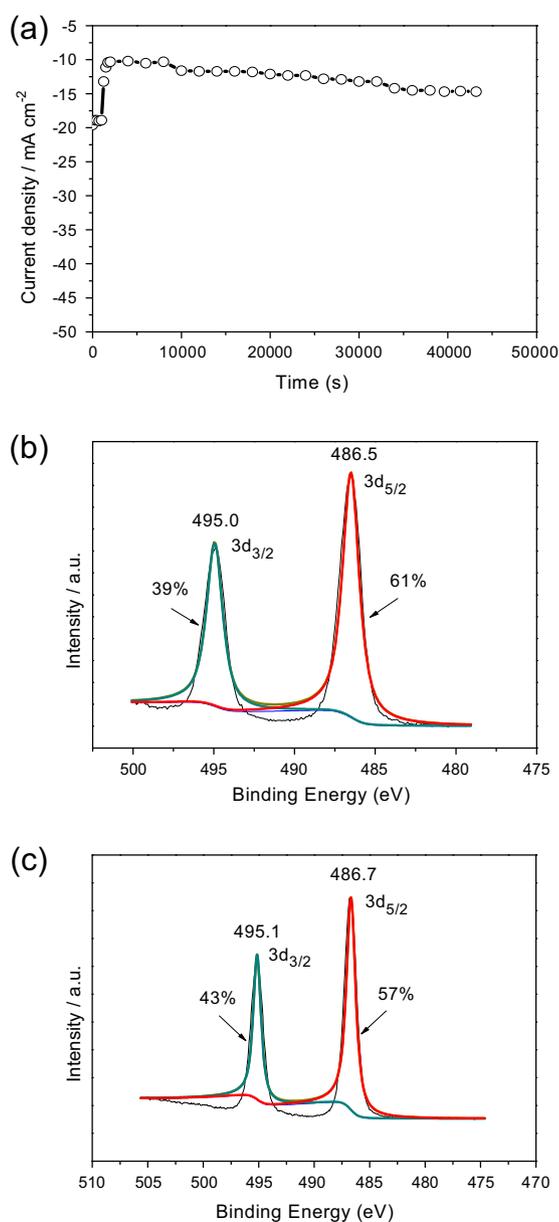
3d<sub>3/2</sub> and Sn 3d<sub>5/2</sub> peaks also exhibit the known characteristics of SnO<sub>2</sub>. As shown in Fig. 7b, the high-resolution Sn3d XPS spectrum before electrolysis exhibits binding energies of 495.0 and 486.5 eV that can be assigned to Sn3d<sub>3/2</sub> and Sn3d<sub>5/2</sub> ionizations, respectively, and the ratio of Sn 3d<sub>3/2</sub>:Sn 3d<sub>5/2</sub> is 39:61. These energies are consistent with Sn(IV) bound to oxygen in SnO<sub>2</sub>. After electrolysis, the binding energies of Sn 3d<sub>3/2</sub> and Sn 3d<sub>5/2</sub> with a fluctuation of only 0.1–0.2, exhibit a Sn 3d<sub>3/2</sub>:Sn 3d<sub>5/2</sub> ratio of 43:57 (Fig. 7c). This suggests there is no occurrence of Sn(IV) oxide to Sn(0) reduction, and the native SnO<sub>2</sub> layer should be stable under the reduction conditions [21]. During electrolysis, the current is sustained without decreasing for at least 12 h at a current density of  $12 \text{ mA cm}^{-2}$  (Fig. 7a), which shows the high durability of the SnO<sub>2</sub>/GDE-86 electrode for CO<sub>2</sub> electroreduction.

## 4. Conclusions

This work presents new experimental results on the development of micro-structured tin oxide catalysts coated GDE electrodes for CO<sub>2</sub> electroreduction in aqueous solution. The morphology- and size-controlled hierarchical tin oxide microspheres (HMS-SnO<sub>2</sub>) are successfully synthesized using a hydrothermal self-assembled process, and employed as the cathode catalysts for CO<sub>2</sub> electroreduction to formate, a form of formate fuel, which can be transported readily or regenerated to electricity using fuel cells. Effects of different proportions of ethanol to distilled water in the synthetic solutions on the catalyst activity and selectivity are investigated. The results show that the catalytic activity is increased with increasing ethanol content from 30% to 86%, indicated by both the increased current density and positive onset potential. Among all electrodes studied, the SnO<sub>2</sub>/GDE-86 electrode can give the most positive onset potential (about  $-0.49 \text{ V}$  vs. SHE). This improvement is further demonstrated by the enhanced Faradaic efficiency observed during electrolysis experiments. The measurement results also show CO<sub>2</sub> reduction efficiencies on these HMS-SnO<sub>2</sub> catalysts are strongly dependent on the catalyst surface structure, size and the composition. The maximum Faradaic efficiency toward formate formation occurs on the SnO<sub>2</sub>/GDE-86 electrode and can be as high as 62% at  $-1.7 \text{ V}$  vs. SHE. In addition, the KHCO<sub>3</sub> electrolyte in the absence of CO<sub>2</sub> is found to show some contribution to the formate production. However, its Faradaic efficiency toward formate production is only 2.4%, which is attributed to the equilibrium between HCO<sub>3</sub><sup>-</sup> and CO<sub>2</sub>. The stability test indicates a high durability of the prepared HMS-SnO<sub>2</sub>/GDE electrode with a stable current density ( $\sim 12 \text{ mA cm}^{-2}$ ) over 12 h of continuous electrolysis operation. XPS spectra analysis demonstrates the SnO<sub>2</sub> layer is stable under the reduction conditions after a long time of electrolysis. The superior performance is credited to the morphology- and a size-controlled hierarchical structure, which may provide more active sites to accelerate the slow kinetics of the CO<sub>2</sub> reduction.

## Acknowledgments

This work was financially supported by the National Natural Science Foundation of China (U1510120), the Project of Introducing Overseas Intelligence High Education of China (2016), the Innovation Program of the Shanghai Municipal Education Commission (14ZZ074), the International Academic Cooperation and Exchange Program of Shanghai Science and Technology Committee (14520721900) and the College of Environmental Science and Engineering, State Environmental Protection Engineering Center for Pollution Treatment and Control in Textile Industry, Donghua University. All the financial supports are gratefully acknowledged.



**Fig. 7.** (a) Stability, and High-resolution Sn 3d XPS spectra of micro-SnO<sub>2</sub> before (b) and after (c) electrolysis for 12 h during the electrochemical reduction of CO<sub>2</sub> in  $0.5 \text{ M KHCO}_3$  solution at  $-1.7 \text{ V}$  vs. SHE.

## Appendix A. Supplementary material

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.apenergy.2016.03.115>.

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